## WHISKER-REINFORCED CERAMIC COMPOSITES FOR

#### HEAT ENGINE COMPONENTS\*

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#### ABSTRACT

Much work has been undertaken to develop techniques of incorporating SiC whiskers into either a  $\mathrm{Si}_3\mathrm{N}_4$  or SiC matrix. The result has been the fabrication of ceramic composites with ever-increasing fracture toughness and strength. To complement this research effort, the fracture behavior of whisker-reinforced ceramics is studied so as to develop methodologies for the analysis of structural components fabricated from this toughened material. The results, outlined herein, focus on the following areas: the use of micromechanics to predict thermoelastic properties, theoretical aspects of fracture behavior, and reliability analysis.

<sup>\*</sup>Work performed on-site at NASA Lewis for the Structural Integrity Branch under grant NCC3-81.

#### SCOPE

Current research activities are being focused on upgrading the performance of heat engines by increasing operating temperatures. With this goal in mind, whisker reinforced ceramics with improved strength and fracture toughness are being investigated for use as structural components in engine hot sections. Even though the fracture toughness of these composites is improved relative to unreinforced ceramics, they remain brittle in nature. Work has been initiated to identify and develop probabilistic methods required in the analysis of whisker-reinforced ceramic components. In addition, approaches using micromechanics to predict thermoelastic properties are under review.

- OBJECTIVES
- TOUGHENING MECHANISMS
- CRACK GROWTH MITIGATION PROCESSES IN WHISKER-REINFORCED CERAMICS
- IMPROVEMENTS IN STRENGTH AND TOUGHNESS
- MICROMECHANICS AND WHISKER OBJENTATION
- ELASTIC MATERIAL PROPERTIES
- WEIBULL DATA
- RELIABILITY ANALYSIS
- SUMMARY

#### OBJECTIVE

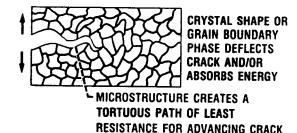
The objective of this effort includes the identification of whisker reinforced ceramics for use in engine hot sections. Further, the technical thrust is directed towards developing and/or refining analytical methods and computer codes that adequately predict fast fracture and life. The effort complements concurrent research by other organizational groups within NASA, the federal government, academia, and industry.

- IDENTIFY WHISKER REINFORCED CERAMICS FOR USE IN ENGINE HOT SECTIONS
- DEVELOP AND REFINE ANALYTICAL METHODS AND COMPUTER CODES FOR PREDICTING
  - FAST FRACTURE
  - LIFE

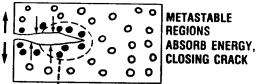
## TYPICAL TOUGHENING MECHANISMS FOR CERAMICS

Historically, three approaches have been taken to improve fracture toughness. One approach is to engineer the grain size and shape to provide a tortuous microstructure such that the path of the crack tip is deflected from the optimum orientation for crack growth. The second approach creates microstructures containing second-phase particles resulting in transformation toughening. Here, a zone surrounding the crack tip absorbs energy and shields the tip by reducing the near field stress. The third approach, including whiskers in the matrix, increases toughness by pinning, deflecting, and/or bridging the crack tip.

#### • ENGINEERING GRAIN SIZE AND SHAPE







LADVANCING CRACK TIP TRANSFORMS
REGIONS AS IT PASSES

#### ADDING HIGH-STRENGTH WHISKERS



WHISKERS ABSORB ENERGY IN DEFLECTION AND/OR PULLOUT

-ADVANCING CRACK TIP BRANCHES, WHISKERS BREAK, TRAILING WHISKERS RESIST OPENING

## CRACK GROWTH MITIGATION PROCESSES - WHISKER REINFORCED CERAMICS

It has been demonstrated experimentally that dispersing whiskers in a brittle matrix will mitigate crack growth. The presence of whiskers at the crack tip modifies fracture behavior by effectively increasing the required crack driving force through several mechanisms. As noted previously, these mechanisms include crack deflection, crack pinning, and whisker bridging. Faber and Evans (1983) have studied crack deflection and provide a lucid discussion. Lange (1971) discussed the process of crack pinning, and Wetherhold (1987) provides a probabilistic treatment of the crack bridging phenomenon.

## THREE PROCESSES INCREASE FRACTURE TOUGHNESS

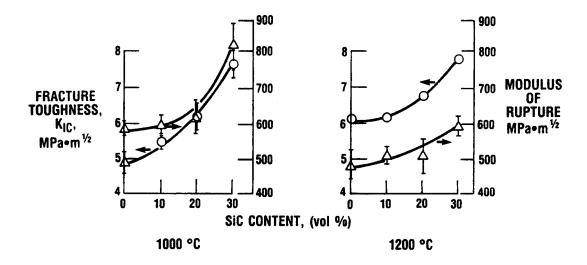
- CRACK DEFLECTION
  - TILTING
  - TWISTING
- CRACK PINNING
- WHISKER BRIDGING

SEVERAL PROCESSES MAY OPERATE SIMULTANEOUSLY

#### IMPROVEMENTS IN STRENGTH AND TOUGHNESS

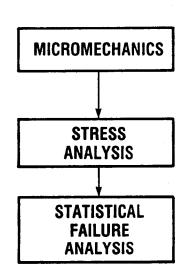
The addition of SiC whiskers in a Si<sub>3</sub>N<sub>4</sub> matrix offers potential for considerable improvement in fracture toughness and strength. Note that the raw materials necessary for fabricating these composites are nonstrategic and are inherently lightweight. Initial attempts to develop whisker composites with these materials met with varying degrees of success (e.g., increased fracture toughness and decreased strength). Recently, Buljan et al. (1987) at GTE Laboratories, Inc., reported improvements in both toughness and strength over the entire range of whisker contents tested.

## HOT PRESSED Si<sub>3</sub>N<sub>4</sub>—SiC WHISKER COMPOSITES



#### ANALYTICAL APPROACH

The analytical approach taken involves the development of an integrated computer algorithm. The algorithm consists of (1) a micromechanics preprocessor, (2) a fintie element code capable of incorporating material anisotropy, and (3) a statistical failure postprocessor. Proposed use of MSC/Nastran as the finite element code is based on its anisotropic analysis capability and widespread user base. The preprocessors and postprocessors are currently under development.



DEVELOP PREPROCESSOR FOR PREDICTION OF ELASTIC THERMOMECHANICAL PROPERTIES

USE EXISTING FINITE-ELEMENT CODES FOR ANISOTROPIC ELASTIC ANALYSIS

DEVELOP AND REFINE FAILURE CRITERIA AND INCORPORATE INTO A POSTPROCESSOR

#### MICROMECHANICS AND WHISKER ORIENTATION

The thermoelastic properties of ceramic whisker composites are determined by process-induced whisker orientation and the constituent properties. Possible material symmetries resulting from fabrication include isotropy, transverse isotropy, and orthotropy. To characterize the internal structure, a whisker orientation distribution function is adopted. This approach was suggested by Pipes et al. (1982). The function quantifies all states of orientation, from random to perfectly aligned.

#### **MATERIAL SYMMETRIES**

- ISOTROPIC
- TRANSVERSELY ISOTROPIC
- ORTHOTROPIC

DEFINE WHISKER ORIENTATION PROBABILITY FUNCTION  $n(\phi)$  SUBJECT TO

$$\int_{-\pi}^{\pi} n(\phi) \ d\phi = 1$$

WHEN

 $n(\phi) = CONSTANT$ 

RANDOM WHISKER ORIENTATION

 $\Pi(\phi) = \delta(\eta - \phi)$ 

**ALIGNED WHISKER ORIENTATION** 

HERE  $\delta$  is the dirac delta function and  $\eta$  is the principal material direction

#### MICROMECHANICS THEORY

The micromechanics theory proposed by Wu and McCullough (1977) for short fiber polymer composites is being considered for adaptation to whisker-reinforced ceramics. The theory was developed using variational techniques and incorporates the degree of whisker orientation, constituent properties, and volume fractions. The method requires the properties of a reference orientation, usually taken as the random orientation state. The parameters f and g define the orientation distribution and are included in the calculation of volume averaged properties.

## EXPRESSING THE EFFECTIVE ELASTIC MATERIAL CONSTANTS AS

$$\langle C_{ijkl} \rangle = C_{ijkl}^{\circ} + F(C_{ijkl}^{\circ}, \overline{C_{ijkl}}) f + G(C_{ijkl}^{\circ}, \overline{C_{ijkl}}) g$$

#### WHERE

 $\langle c_{iikl} 
angle$  predicted elastic constants of the composite

Ciin Elastic Constants for Random Whisker Orientation

 $\overline{\textit{\textbf{C}}_{iikl}}$  linear combination of the constituent elastic constants

THE PARAMETERS f AND g ARE FUNCTIONALLY DEPENDENT UPON  $n(\phi)$  AND DESCRIBE THE ORIENTATION STATE. WHEN

f=g=0 RANDOM WHISKER ORIENTATION

f=g=1 ALIGNED WHISKER ORIENTATION

#### PLANAR WHISKER ORIENTATION

The planar orientation of whiskers is often encountered in hot pressed composites. This special case was considered by Pipes et al. (1982). The orientation descriptors  $f_p$  and  $g_p$  along with the distribution function are defined below. Bozarth, et al. (1987) developed a Monte Carlo simulation that depicts the whisker orientation for specified values  $f_p$ .

## APPLYING THE CONCEPT TO A PLANAR ORIENTATION OF WHISKERS

$$n(\phi) = K \cos(\lambda \phi)$$

$$f_D = 2\langle \cos^2 \phi \rangle - 1$$

$$g_{p} = \frac{2f_{p}(7 - 2f_{p})}{5(4 - 2f_{p})}$$

**WHERE** 

$$\langle \cos^2 \phi \rangle = \int_{-\pi/2}^{\pi/2} n(\phi) \cos^2 \phi \, d\phi$$

THEN

 $f_p = 0.1$  TRANSVERSE ISOTROPY

 $0 < f_p < 1$  ORTHOTROPY

## A MONTE CARLO SIMULATION ILLUSTRATES GRAPHICALLY VARIOUS $f_{ ho}$ VALUES



 $f_{n} = 0.0$ 



 $f_p = 0.3$ 



 $f_{p} = 0.6$ 

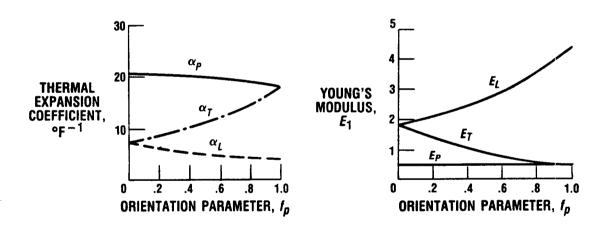


 $f_{p} = 0.9$ 

#### INFLUENCE OF WHISKER ORIENTATION ON MATERIAL PROPERTIES

The influence of orientation on material properties is show below (from Pipes et al., 1982). Both the elastic modulii and the coefficients of thermal expansion are plotted for a short glass fiber phenolic resin matrix composite. The constants are plotted over the full range of  $f_p$ . Note that for  $f_p=1$  and  $f_p=0$ , the material is transversely isotropic. For all other values, the material is orthotropic.

# PREDICTIONS USING THE APPROACH FOR A SHORT FIBER POLYMER COMPOSITE FIBER ASPECT RATIO $\simeq 40-100$ ; FIBER CONTENT BY WEIGHT = 58%

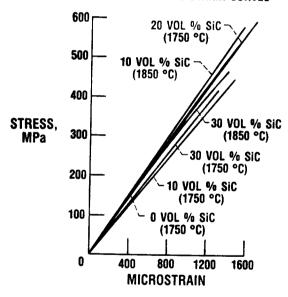


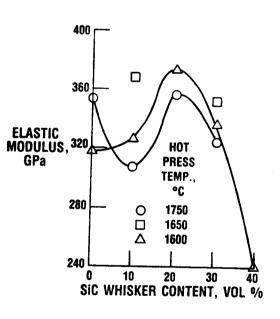
## ELASTIC MATERIAL PROPERTIES

The examples shown below of stress-strain curves corresponding to various whisker contents were reported by Shalek et al. (1986). Note that the curves exhibit a linear stress-strain response up to the point of fracture. This indicates that crack growth involves only brittle fracture and allows for the application of linear elastic fracture mechanics (LEFM).

## SIC WISKER-HOT-PRESSED Si3N4

## FOUR-POINT BEND STRESS-STRAIN CURVES



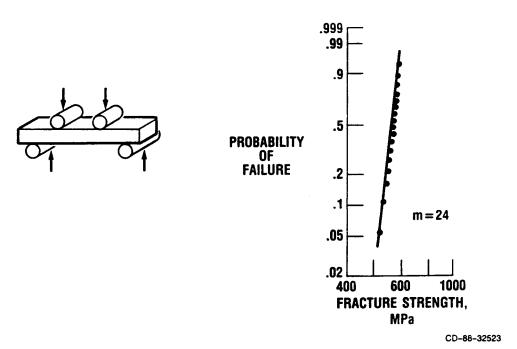


#### WEIBULL MODULUS

Greatly improved reliability has been cited by a number of authors, including Claussen and Petzow (1985), who have reported the highest Weibull modulus (m = 24) in the open literature. Improved processing techniques have resulted in the reduction of inhomogeneities, uniform whisker distribution, and a dense matrix. However, the variability of strength is still too high for the application of deterministic fracture theories.

## 30% SiC-WHISKER-Si3N4-MATRIX COMPOSITE

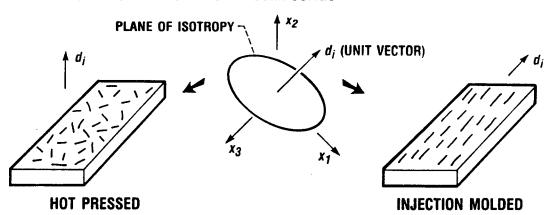
WHISKER REINFORCED CERAMICS EXHIBIT A VARIABILITY IN STRENGTH; HENCE, PROBABILISTIC METHODS OF ANALYSIS MUST BE APPLIED



#### TRANSVERSE ISOTROPY

As noted previously, depending on fabrication, a whisker composite may have isotropic, transversely isotopic, or orthotropic material symmetry. It is anticipated that the probabilistic methods used in the analysis of monolithic ceramics will be appropriate for isotropic whisker composites. However, for transversely isotropic whisker composites, the probability of failure  $P_f$  must also reflect the preferred direction of the material  $d_i$ . This direction is defined as the normal to the plane of isotropy. A simple but rigorous method of including material symmetry in the functional dependence of  $P_f$  for transverse isotropy is presented. A similar approach can be developed for orthotropy.

#### TRANSVERSELY ISOTROPIC WHISKER COMPOSITES



DEPENDENCE OF  $P_f$  MUST REFLECT  $\sigma_{ij}$ ,  $d_i$ , volume as well as the weibull parameters

$$P_f = P_f (\sigma_{ij}, d_i d_j, V, ....)$$

ASSUMING WEAKEST LINK THEORY IS APPROPRIATE, FORMULATE  $P_f$  IN A WAY THAT ACCOUNTS FOR THE ABOVE DEPENDENCE

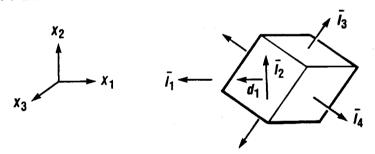
## TENSORIAL INVARIANTS

As Pf is a scalar function, it must remain form invariant under arbitrary proper orthogonal transformations. Form invariance is ensured if dependence is taken on invariants that constitute an integrity basis or any subset thereof. Tensorial invariant theory (see Spencer (1971)) serves as the basic mathematical tool in the development of the integrity basis. A subsequent geometric argument is made in constructing a slightly different set of invariants that corresponds to physical mechanisms related to fracture.

## ADOPT THE FOLLOWING INTEGRITY BASIS:

$$\begin{split} I_1 &= \sigma_{ii} & I_4 &= d_i \ d_j \ \sigma_{ij} \\ I_2 &= \sigma_{ij} \ \sigma_{ji} & I_5 &= d_i \ \sigma_{ij} \ \sigma_{jk} \ d_k \\ I_3 &= \sigma_{ij} \ \sigma_{jk} \ \sigma_{ki} & \end{split}$$

## IDENTIFY DAMAGING STRESS TRACTION VECTORS:



- $ar{I_3}$ ,  $ar{I_4}$  maximum and minimum normal stresses in plane of isotropy
- $\bar{l}_1$  NORMAL STRESS IN DIRECTION OF  $d_i$
- $ar{\it l}_2$  SHEAR STRESS ACTING ACROSS  $\it d_i$

#### FAILURE PROBABILITY AS A FUNCTION OF INVARIANTS

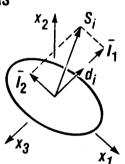
The dependence of  $P_f$  is subsequently taken on the invariants  $\bar{I}_1$ ,  $\bar{I}_2$ ,  $\bar{I}_3$ , and  $\bar{I}_4$ . This new set of invariants corresponds to the magnitudes of the stress traction vectors assumed to be the primary causes of fracture. These invariants incorporate both the stress tensor  $\sigma_{ij}$  and the direction vector  $d_i$ . Adopting the simplest of probabilistic failure theories, that is, a noninteractive theory, results in the form of  $P_f$  shown in the figure.

FROM THE INTEGRITY BASIS CONSTRUCT INVARIANTS WITH PHYSICAL INTERPRETATIONS CORRESPONDING TO THE MAGNITUDES OF THE DAMAGING STRESS TRACTION VECTORS

$$ar{I_1} = I_4$$
= COMPONENT OF  $S_i$ 
PROJECTED ON
DIRECTION  $d_i$ ,

$$\bar{l}_3 = \frac{1}{2}(l_1 - l_4) + [(\frac{1}{2})l_2 - l_5] + \frac{1}{4}(l_4^2 - l_1^2) + (\frac{1}{2})l_1 l_4]^{\frac{1}{2}}$$

= MAXIMUM NORMAL STRESS IN PLANE OF ISOTROPY



 $S_i = \sigma_{ii}d_i$ 

$$\bar{l}_2 = (l_5 - l_4^2)^{V_2}$$
= COMPONENT OF  $S_i$ 
PROJECTED ON THE
PLANE OF ISOTROPY

$$\bar{l}_4 = \frac{1}{2}(l_1 - l_4) - \left[\frac{1}{2}l_2 - l_5\right] + \frac{1}{4}(l_4^2 - l_1^2) + \frac{1}{4}2l_1l_4$$

= MAXIMUM NORMAL STRESS IN PLANE OF ISOTROPY

**TAKING** 

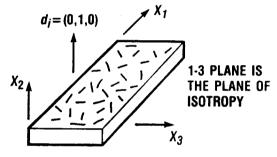
$$P_{f} = P_{f} (\bar{l}_{1}, \bar{l}_{2}, \bar{l}_{3}, \bar{l}_{4}, V, ...)$$

ENSURES  $P_I$  IS FORM INVARIANT. ASSUMING THE INVARIANTS ACT SEPARATELY IN PRODUCING FAILURE

$$P_{I} = 1 - \exp \left\{ - \int_{V} \left[ \left( \frac{\bar{l}_{1}}{\beta_{1}} \right)^{\alpha_{1}} + \left( \frac{\bar{l}_{2}}{\beta_{2}} \right)^{\alpha_{2}} + \left( \frac{\bar{l}_{3}}{\beta_{3}} \right)^{\alpha_{3}} + \left( \frac{\bar{l}_{4}}{\beta_{3}} \right)^{\alpha_{3}} \right] dV \right\}$$

#### PLANE STRESS PROBABILISTIC ANALYSIS

The form of  $P_f$  is simplified for plane stress conditions considering two planar orientations of the whiskers. In both cases it is assumed that the whiskers are confined to the 1-3 plane due to fabrication. A random orientation of whiskers in the 1-3 plane reduces  $P_f$  to the isotropic formulation. Alternatively, perfect alignment of the whiskers in the 1-3 plane reduces  $P_f$  to a formulation proposed by Sun and Yamada (1978), Wetherhold and Pipes (1984), and Cassenti (1984).

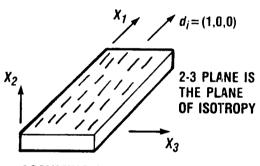


ASSUMING PLANE STRESS  $\sigma_{12} = \sigma_{22} = \sigma_{23} = 0$ 

THEN

$$P_{1} = 1 - \exp \left\{-\int_{V} \left[\left(\frac{\sigma_{1}}{\beta_{3}}\right)^{\alpha_{3}} + \left(\frac{\sigma_{2}}{\beta_{3}}\right)^{\alpha_{3}}\right] dV\right\} \qquad P_{1} = 1 - \exp \left\{-\int_{V} \left[\left(\frac{\sigma_{11}}{\beta_{1}}\right)^{\alpha_{1}} + \left(\frac{\sigma_{12}}{\beta_{2}}\right)^{\alpha_{2}}\right] dV\right\}$$

WHERE  $\sigma_1$  AND  $\sigma_2$  ARE THE PRINCIPLE STRESSES IN THE PLANE OF ISOTROPY



ASSUMING PLANE STRESS  $\sigma_{13} = \sigma_{23} = \sigma_{33} = 0$ 

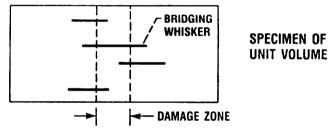
THEN

$$P_{1} = 1 - \exp \left\{ - \left| \left| \left( \frac{\sigma_{11}}{\beta_{1}} \right)^{\alpha_{1}} + \left( \frac{\sigma_{12}}{\beta_{2}} \right)^{\alpha_{2}} + \left( \frac{\sigma_{22}}{\beta_{3}} \right)^{\alpha_{3}} \right| dV \right\}$$

#### WHISKER BRIDGING

Attempts have been made to develop statistical models that account for the microstructural events leading to crack propagation. Wetherhold (1987) derived a model assuming that fracture behavior is dominated by whiskers bridging a critical damage zone. The damage zone is analogous to a microcrack which is expected to grow and coalesce with other microcracks during progressive fracture. A distribution function for composite strength is developed based on the incorporation of random whisker strength into a bundle fracture theory.

# THE FOLLOWING STATISTICAL APPROACH ACCOUNTS FOR THE WHISKER BRIDGING MECHANISM



**DEFINE** 

[A] = THE EVENT WHERE ULTIMATE COMPOSITE STRENGTH IS GREATER THAN THE APPLIED LOAD

THEN

$$P_S = P[A]$$

$$= \sum_{i=0}^{N} P[A|n=i] P[n=i]$$

WHERE

**n** NUMBER OF WHISKERS BRIDGING THE DAMAGE ZONE

**N** NUMBER OF FIBERS PER UNIT VOLUME

#### SUMMARY

Enough experimental data exist that strongly suggests whisker-reinforced ceramics have promise, especially in the automotive industry where low-cost, high-volume fabrication is a necessity. The Structural Integrity Branch is identifying and refining existing design methodologies and, where necessary, assisting the research community in developing new methodologies. At present, there exists a need for analytical methods that capture the microstructural events that lead to increased fracture toughness and strength. Finally, work has begun on an integrated computer program capable of predicting elastic material properties through the application of micromechanics, the state of stress within a structural component, and the reliability of the component given the state of stress.

- EXPERIMENTAL DATA SHOW SIGNIFICANT INCREASES OF STRENGTH AND FRACTURE TOUGHNESS DUE TO THE ADDITION OF CERAMIC WHISKERS IN A CERAMIC MATRIX
- WHISKER COMPOSITES HAVE THE ATTRACTIVE FEATURE OF USING CONVENTIONAL POWDER PROCESSING TECHNIQUES IN HIGH-VOLUME, LOW-COST FABRICATION
- AN INTEGRATED STRUCTURAL ANALYSIS CODE IS BEING DEVELOPED WITH THE FOLLOWING COMPONENTS:
  - MICROMECHANICS PREPROCESSOR
  - FINITE-ELEMENT PROGRAM
  - STATISTICAL FAILURE ANALYSIS POSTPROCESSOR
- RESEARCH IS UNDERWAY TO DEVELOP AND REFINE STATISTICAL FAILURE THEORIES
   TO ACCOUNT FOR MATERIAL ANISOTROPY ALONG WITH THE MICROSTRUCTURAL
   EVENTS LEADING TO FAILURE (I.E., CRACK DEFLECTION AND WHISKER BRIDGING)

#### REFERENCES

- Bozarth, M.J., Gillespie, J.W., and McCullough, R.L., 1987, "Fiber Orientation and Its Effect Upon Thermoelastic Properties of Short Carbon Fiber Reinforced Poly (etheretherketone) (PEEK), "Polymer Composites, Vol. 8, pp. 74-81.
- Buljan, S.T., Baldoni, J.G., and Huckabee, M.L., 1987, "Si<sub>3</sub>N<sub>4</sub>-SiC Composites," Ceramic Bulletin, Vol. 66, pp. 347-352.
- Cassenti, B.N., 1984, "Probabilistic Static Failure of Composite Material," AIAA Journal, Vol. 22, pp. 113-111.
- Claussen, H., and Petzow, G., 1985, "Whisker-reinforced Zirconia Toughened Ceramics," Tailoring Multiphase and Composite Ceramics; Proceedings of the 21st University Conference on Ceramic Science, Plenum Press, New York, pp. 649-662.
- Faber, K.T., and Evans, A.G., 1983, "Crack Deflection Processes I. Theory," <a href="Acta Metallurgica">Acta Metallurgica</a>, Vol. 31, pp. 565-576,
- Lange, F.F., 1971, "The Interaction of a Crack Front with a Second-phase Dispersion," <a href="Philosophy Magazine">Philosophy Magazine</a>, Vol. 22, pp. 983-992.
- Pipes, R.B., McCullough, R.L., and Taggart, D.G., 1982, "Behavior of Discontinuous Fiber Composites: Fiber Orientation," <u>Polymer Composites</u>, Vol. 3, pp. 34-39.
- Shalek, P.D., Petrovic, J.J., Hurley, G.F., and Gac, F.D., 1986, "Hot-Pressed SiC/Si<sub>3</sub>N<sub>4</sub> Matrix Composites," Ceramic Bulletin, Vol. 65, pp. 351-356.
- Spencer, A.J.M., 1971, "Theory of Invariants," <u>Continuum Physics</u>, Vol. I, Eringen, A.C., ed., Academic Press, London, 1971.
- Sun, C.T., and Yamada, S.E., 1978, "Strength Distribution of a Unidirectional Fiber Composite," Journal of Composite Materials, Vol. 12, pp. 169-176.
- Wetherhold, R.C., 1987, "Probabilisitic Aspects of the Strength of Fiber-dominated Short-fiber Composites I: Aligned Fibers," <u>Materials Science and Engineering</u>, Vol. 97, pp. 7-12.
- Wetherhold, R.C., and Pipes, R.B., 1984, "Statistics of Fracture of Composite Materials Under Multiaxial Loading," <u>Material Science and Engineering</u>, Vol. 68, pp. 113-118.
- Wu, C.T.D., and McCullough, R.L., 1977, "Constitutive Relationships for Heterogeneous Materials," <u>Developments in Composite Materials</u>. Hollister, G.S., ed., Applied Science Publishers, Ltd., London, U. K., pp. 119-188.